



Modeling phase transformations in Mn-rich cathodes with charge-informed machine-learning interatomic potentials



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Background

- Mn-rich disordered rocksalt cathodes (DRX) materials are promising **earth-abundant** materials for Li-ion batteries, with the potential to scale sustainable energy storage solutions to TWh/year.
- The degree of **cation orderings** can change upon battery cycling, which affects the electrochemistry (e.g., voltage profile) [1].
- There are significant challenges in modeling the structure vs. electrochemistry from *ab-initio* considering their **compositional complexity**, substantial **site disorder** and the state with **partial disorder** with quantum accuracy.

Methodology

- Cluster expansion** as a lattice model to capture the cation-disordered states. The CE casts the energy from *ab-initio* calculation as a function of the occupancy of atoms on a set of predefined sites [2, 3]:

$$E(\sigma) = \sum_{\beta} m_{\beta} J_{\beta} \langle \Phi_{\alpha \in \beta} \rangle_{\beta} + \frac{E_0}{\epsilon_r}, \quad \Phi_{\alpha} = \prod_{i=1}^N \phi_{\alpha_i}(\sigma_i). \quad (1)$$

- CHGNet** is a graph neural network (GNN) based machine-learning interatomic potential (MLIP) [4]:

$$E_{\text{tot}} = \sum_i \phi_E(v_i^n), \quad \mathbf{f}_i = -\frac{\partial E_{\text{tot}}}{\partial \mathbf{x}_i}, \quad m_i = \phi_m(v_i^{n-1}) \quad (2)$$

- Potential energy surface samplings with r²SCAN-DFT:

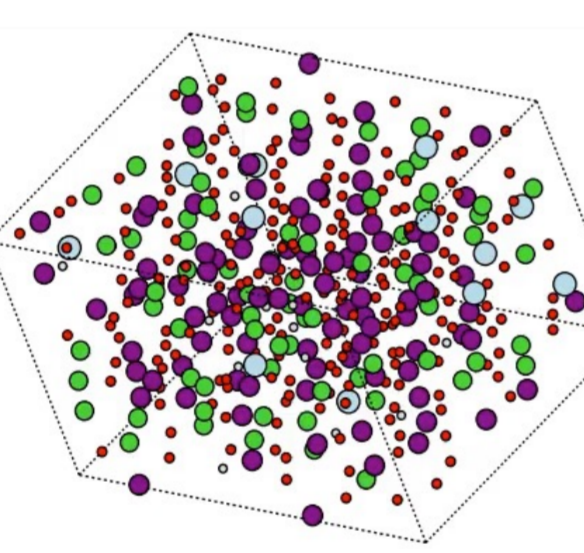
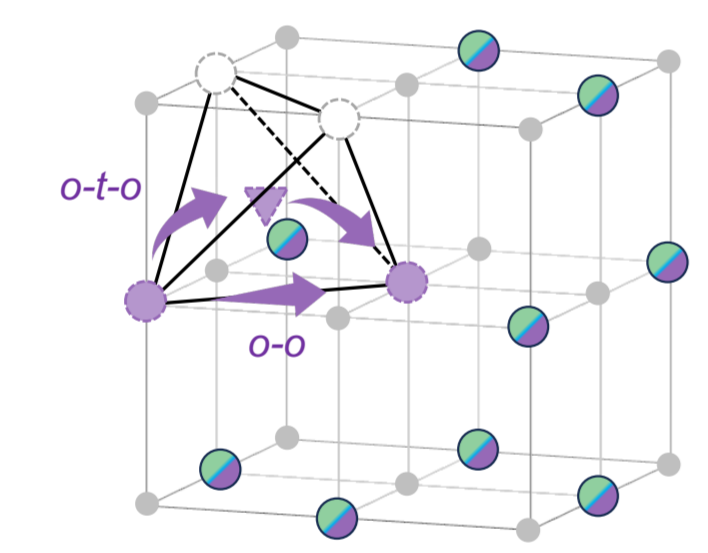
Relaxation from existing datasets (e.g. MP)

Transition states sampling

MLIP-MD Trajectories



ENCUT = 680 eV
EDIFF = 1e-6



- Monte Carlo Simulations** were implemented at $T = 1273$ K to simulate the cation-disordered structure (including short-range order effect) [5]

- Molecular Dynamics** were implemented at $T = 1273$ K for 2 ns to simulate the phase transformation from the DRX to δ -phase (partially disordered + spinel-like ordering)

- Equilibrium voltage profiles** from thermodynamics were evaluated using

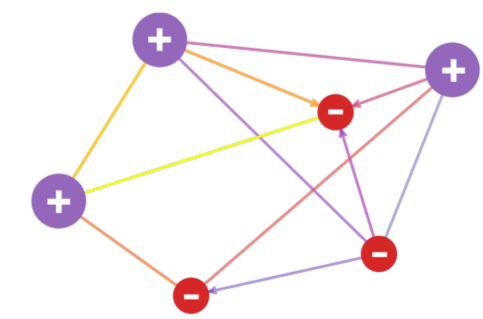
$$\bar{V}(x_1, x_2) \approx -\frac{E_{\text{Li}_{x_1}\text{TMO}_2} - E_{\text{Li}_{x_2}\text{TMO}_2} - (x_1 - x_2)E_{\text{Li}}}{F(x_1 - x_2)} \quad (3)$$

where E_{Li} is the internal energy of the bcc Li metal, $E_{\text{Li}_x\text{TMO}_2}$ represents the internal energy of the (de)lithiated structures.

References



CHGNet



¹Z. Cai et al., Nature Energy **9**, 27–36 (2023).

²P. Zhong et al., PRX Energy **2**, 043005 (2023).

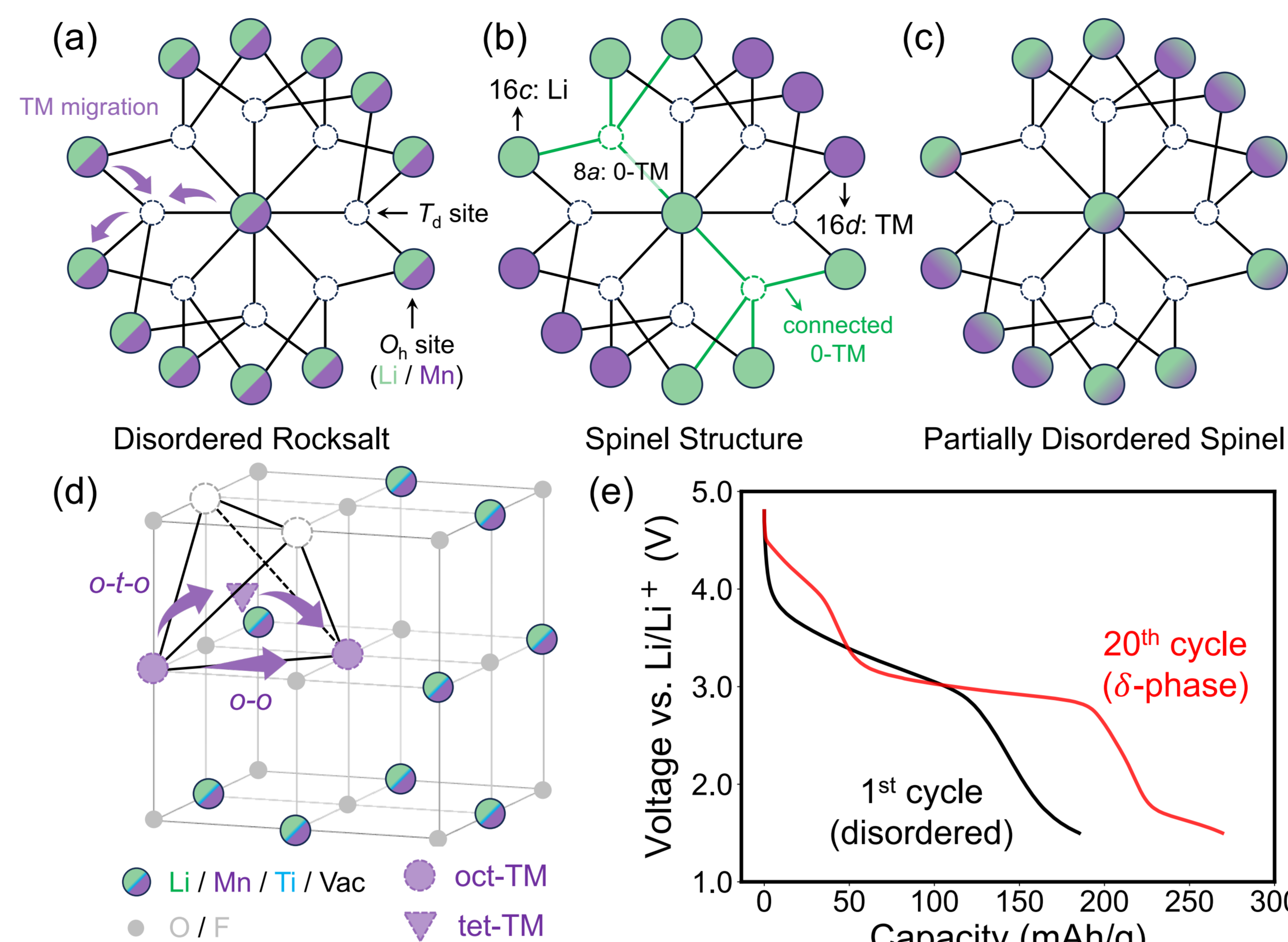
³L. Barroso-Luque et al., Journal of Open Source Software **7**, 4504 (2022).

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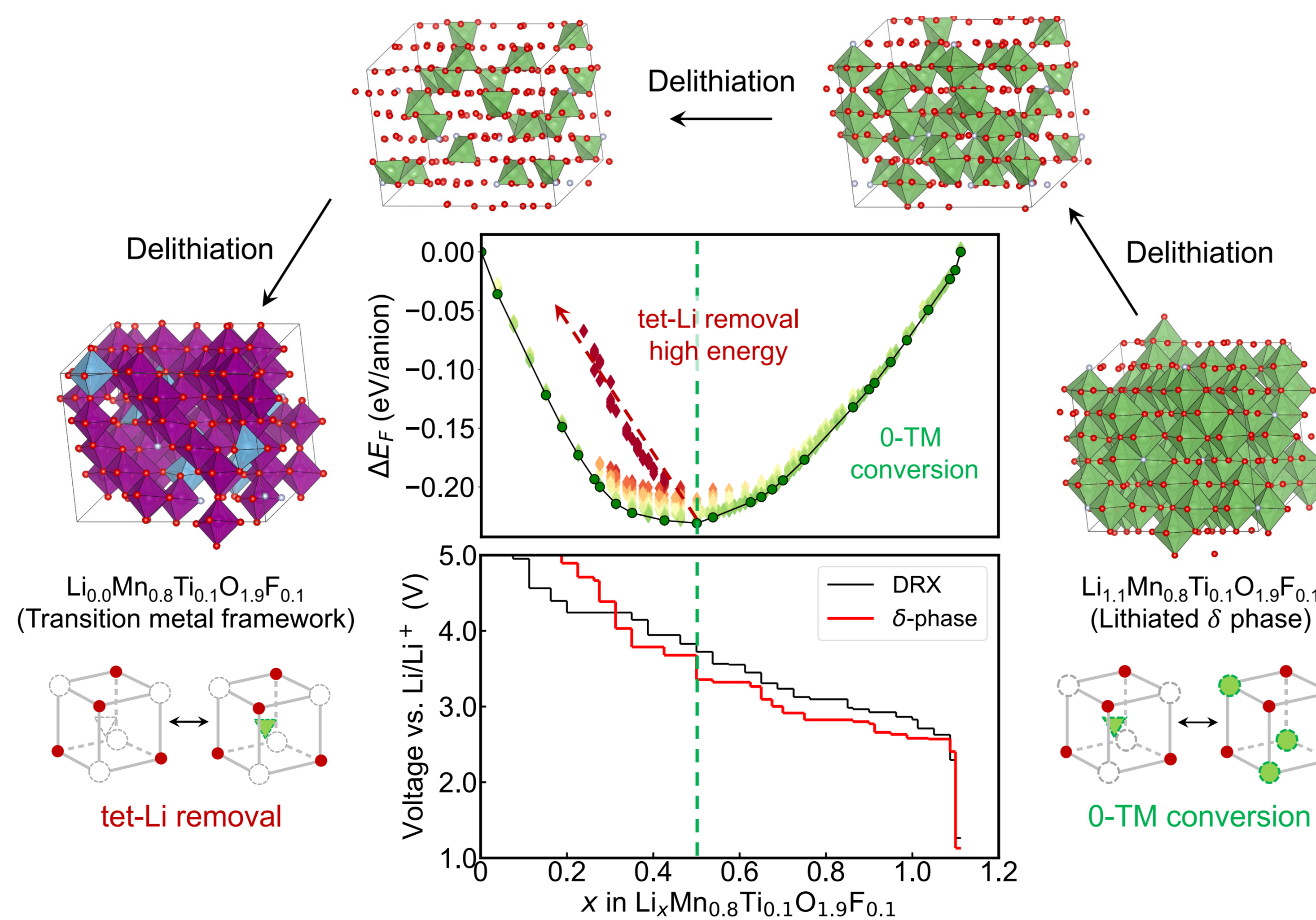
⁵P. Zhong et al., Chemistry of Materials **32**, 10728–10736 (2020).

⁶J. Reed et al., Electrochemical and Solid-State Letters **4**, A78 (2001).

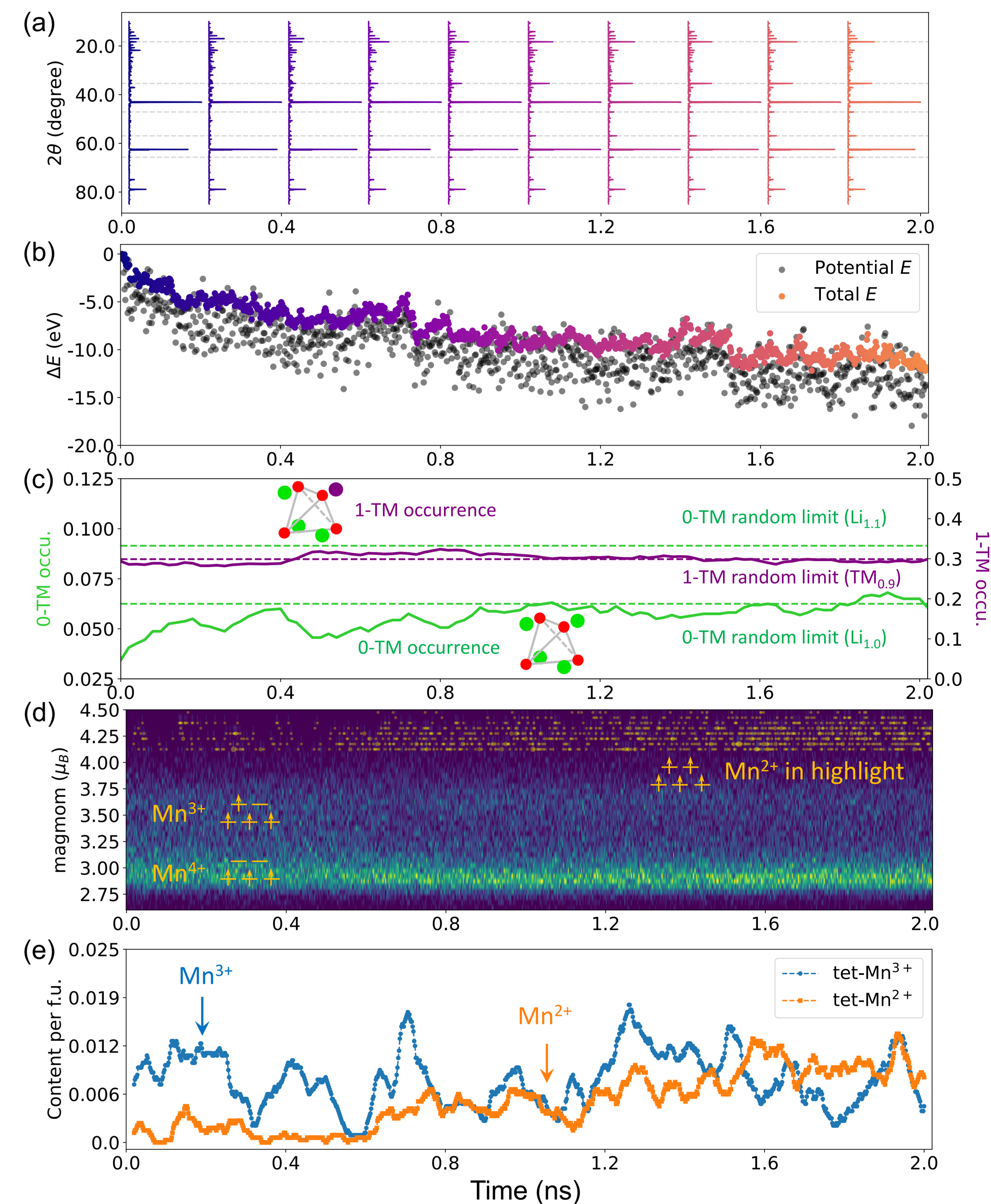
Cation orderings vs. electrochemistry



Equilibrium Intercalation Voltage Profiles



Structure and charge evolution from charge-informed MD simulations



Conclusion

- We successfully modeled the phase transformation in complex oxides using charge-informed MLIP-MD and rationalized the relation between structure change and electrochemistry.
- No two-phase reaction has been observed for the δ -phase, which exhibits (1) systematically lower average voltage around 3 V and (2) high voltage features around 4 V in the charged states compared to that of the DRX phase.
- We discovered that contrary to the previous belief attributing Mn^{2+} ions to phase transformation due to their low migration barriers [6], **the emergence of Mn^{2+} is correlated to the formation of spinel-like ordering** (ns-scale phenomenon) rather than a local hopping (ps-scale). Instead, Mn^{3+} is the initial majority of migrating ions that drive the transformation.